

Reentrant Behavior of the Temperature Dependence of Resistivity of
DCNQI-Cu Alloy System, $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$

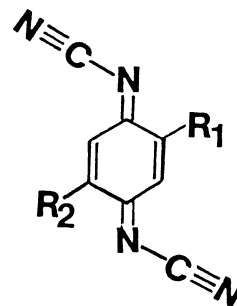
Akiko KOBAYASHI, Reizo KATO,[†] and Hayao KOBAYASHI[†]

Department of Chemistry, Faculty of Science, The University of Tokyo,
Hongo, Bunkyo-ku, Tokyo 113

[†]Department of Chemistry, Faculty of Science, Toho University,
Funabashi, Chiba 274

Reentrant resistivity behavior (metal→semiconductor→metal) is observed at ambient pressure in the alloy system of $[(\text{DMe-DCNQI})_{1-x}(\text{MeBr-DCNQI})_x]_2\text{Cu}$ ($x < 0.1$). Resistivity maximum appears around 60 K. The magnitude of the resistivity maximum is extremely sensitive to x . This anomaly resembles the anomalous pressure-induced reentrant resistivity behavior in $(\text{R,R-DCNQI})_2\text{Cu}$ ($\text{R} = \text{CH}_3, \text{CH}_3\text{O}$).

Recent progress in the field of the molecular conductors has brought about a new interesting system, $(\text{R}_1, \text{R}_2\text{-DCNQI})_2\text{Cu}$ ($\text{R}_1, \text{R}_2 = \text{CH}_3, \text{CH}_3\text{O}, \text{Cl}, \text{Br}, \text{I}$; DCNQI = N,N'-dicyanoquinonediimine). The crystals are composed of the one-dimensional (1D) columns of $\text{R}_1, \text{R}_2\text{-DCNQI}$ molecules interconnected through the tetrahedrally coordinated Cu cations.¹⁻³ They have many exotic features: (1) $2p\pi\text{-}d$ mixing in the formation of the conduction band (2) mixed-valency of Cu ($\text{Cu}^+, \text{Cu}^{2+}$) and antiferromagnetic interaction of Cu^{2+} spins via organic molecules and (3) pressure-induced resistivity anomaly.



$\text{R}_1, \text{R}_2\text{-DCNQI}$

All these features are considered to be interrelated with each other. The DCNQI-Cu systems can be classified into two groups: (I) the system with stable metallic state down to at least 0.5 K (e.g. $(\text{DMe-DCNQI})_2\text{Cu}$, $\text{R}_1 = \text{R}_2 = \text{CH}_3$) and (II) the system with sharp metal-insulator (MI) transition (e.g. $(\text{MeBr-DCNQI})_2\text{Cu}$, ($\text{R}_1 = \text{CH}_3, \text{R}_2 = \text{Br}$) ($T_{\text{MI}} = 156 \text{ K}$)). The group-I compound shows a pressure-induced metal instability above the critical pressure (P_c) and transforms to group-II compound at higher pressure.⁴⁻⁶ In the case of $(\text{DMe-DCNQI})_2\text{Cu}$, P_c is extremely low ($\approx 50 \text{ bar}$). In usual molecular conducting systems with one-dimensional (1D) nature, the MI transitions are originated from the nature of the electric susceptibility ($\chi(q, T)$) of 1D electron gas, which diverges at $q = 2k_F$ and $T = 0 \text{ K}$. Since pressure tends to suppress this divergence, the metallic state becomes more stable under high pressure. Therefore, the pressure-induced metal instability observed in the DCNQI-Cu system with the stacking structure of planar π -molecules is not so trivial. In addition, this system exhibits a unique temperature dependence of resistivity (ρ) at high pressure (metal → semiconductor → metal).^{5,6} For

example, the metallic state of $(\text{DMeO-DCNQI})_2\text{Cu}$ becomes unstable above 8 kbar and below ca. 100 K. The system reenters into the metallic phase below ca. 60 K. We have recently proposed that the DCNQI-Cu system is a possible candidate of the molecular metal which belongs to a family of so-called dense Kondo materials ever known in the metallic compounds of lanthanides and actinoids such as CeB_6 and UPt_3 . The interaction between $2p\pi$ conduction electrons and the magnetic ions (Cu^{2+}) is considered to be of central importance.⁶⁾

In this paper, we will report that similar "reentrant resistivity behavior" is also observable at ambient pressure in the alloy system, $[(\text{DMe-DCNQI})_{1-x}(\text{MeBr-DCNQI})_x]_2\text{Cu}$ (hereafter, $[(\text{DMe-DCNQI})_{1-x}(\text{MeBr-DCNQI})_x]_2\text{Cu}$ is abbreviated to $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$).

The black needle crystals of $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$ were prepared electrochemically from the acetonitrile solutions of DMe-DCNQI and MeBr-DCNQI with various mixing ratios. $[\text{Cu}(\text{CH}_3\text{CN})_4]\text{ClO}_4$ was used as a supporting electrolyte. The values of x were determined on 13 crystals by the X-ray crystal structure refinements of the occupancy probability of the Br-site (or Me-site).⁷⁾ The mixing ratio of DMe-DCNQI and MeBr-DCNQI in the acetonitrile solution was largely different from the DMe/MeBr ratio in the crystal. As shown in Fig. 1, MeBr-DCNQI tends to be highly concentrated in the solid. The electrical resistivities were measured by conventional four probe method over the temperature range of 4.2 - 300 K. Gold wires of 15 μm diameter were bonded to the crystals with gold paint and used as the electrical leads.

The lattice constants of $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$, determined by a Rigaku automated X-ray four-circle diffractometer are not simple arithmetic means of the lattice constants of $(\text{DMe-DCNQI})_2\text{Cu}$ and $(\text{MeBr-DCNQI})_2\text{Cu}$. The cell volume takes a

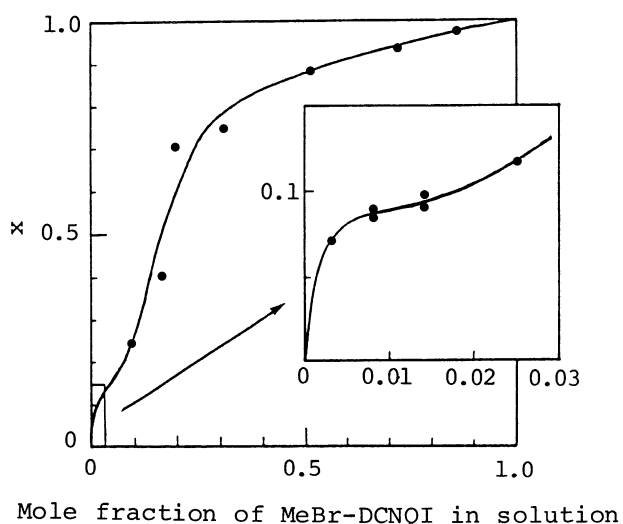


Fig. 1. Correlation diagram between x determined from X-ray crystal structure refinement and the mole fraction of MeBr-DCNQI in solution.

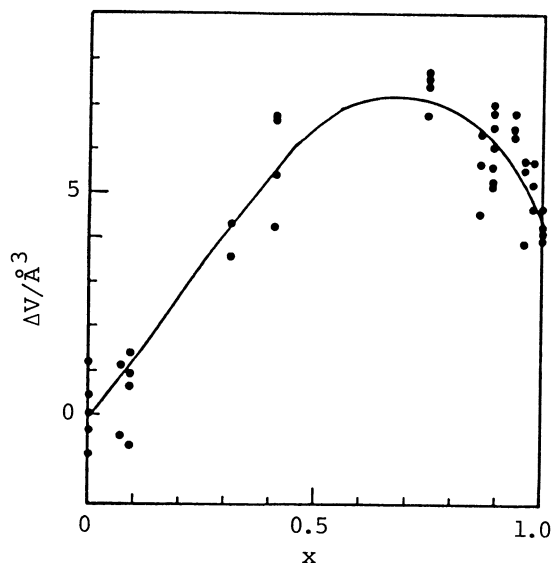


Fig. 2. Unit cell volume of $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$ as a function of x . The cell volume of $(\text{DMe-DCNQI})_2\text{Cu}$ is 1797.4 \AA^3 ($x=0.0$).

maximum around $x = 0.6-0.7$ (Fig. 2). Similar non-additivity also appeared in the temperature dependence of the resistivity. Since $(\text{DMe-DCNQI})_2\text{Cu}$ is metallic down to low temperature and $(\text{MeBr-DCNQI})_2\text{Cu}$ undergoes a MI transition at 156 K, it might be imagined that the alloy system would exhibit a MI transition whose transition temperature decreases continuously with increasing x . However, the observed resistivity behavior was quite different. The modes of the temperature dependences of the resistivity changes can be classified into four types: (I) $x < 0.1$; a reentrant resistivity behavior (metal \rightarrow semiconductor \rightarrow metal) similar to that observed in the group-I compound at high pressure was observed (Fig. 3). The residual resistivity seems to increase rapidly with x , indicating the strong effect of the lattice disorder. The reentrant behavior is accompanied by the hysteresis. The possible existence of some kind of domain structures (or non-uniformity) of the alloy system might be responsible for it. (II) $0.1 < x < 0.6$; the T - ρ curve shows a large hysteresis. In the cooling cycle, the system is metallic down to ca. 90 K where ρ increases almost discontinuously, while the system remains to be low-conductive up to ca. 210 K in the warming cycle. (III) $x \approx 0.7$; the system shows semiconductive behavior for all over the temperature range, which may be ascribed to the electron localization due to the expected large lattice disorder. (IV) $x > 0.8$; the system shows a sharp MI transition around 155 K similar to that of pure $(\text{MeBr-DCNQI})_2\text{Cu}$, where a small hysteresis was observed.

As mentioned above, the increase of ρ followed by the reentrance of the metallic phase has been already found in group-I compound at high pressure.

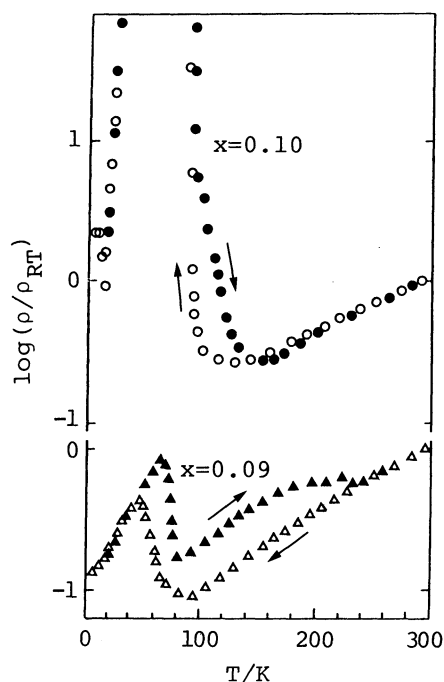


Fig. 3. Temperature dependence of the resistivity of $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$. Room-temperature resistivities are ca. $10^{-3} \Omega \text{ cm}$.

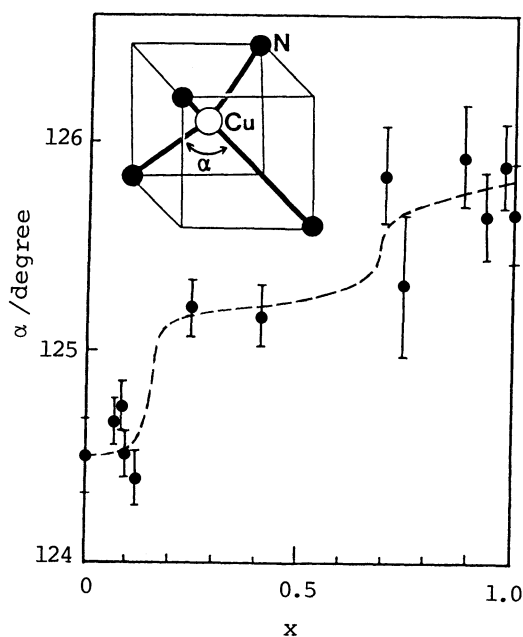


Fig. 4. The x dependence of α .

Therefore, the increase of x and the pressure might be considered to have similar effect on the crystal lattice. However, the situation is not so simple. Since the unit cell volume of $(\text{DMe-DCNQI})_2\text{Cu}$ is smaller than that of $(\text{MeBr-DCNQI})_2\text{Cu}$, the replacement of DMe-DCNQI by MeBr-DCNQI makes the cell volume large. This is contrary to the pressure effect. Therefore, the observation of the reentrant ρ behavior in the alloy system suggests that the reduction of the cell volume is not essential to induce the metal instability. As pointed out before, the magnitude of the $\bar{4}$ distortion of the coordination geometry around Cu seems to be essential for determining the stability of the metallic state of the DCNQI-Cu system.^{2,3)} The angle α ($\angle \text{N-Cu-N}$) can be regarded as a good measure of the magnitude of the crystal field splitting of 3d orbitals of Cu (see Fig. 4). The x dependence of α , determined by the X-ray crystal structure refinement, shows that α increases stepwise with increasing x . There are three regions: small- α region ($x < 0.1$), intermediate α -region ($0.1 < x < 0.6$) and large- α region ($0.8 < x$). The reentrant resistivity behavior was observed in the small- α region, where the coordination geometry is almost identical to that of pure $(\text{DMe-DCNQI})_2\text{Cu}$. This is consistent with that P_c of $(\text{DMe-DCNQI})_2\text{Cu}$ is extremely low (≈ 50 bar).

In summary, we have observed the metal instability and the reentrant resistivity behavior in the DCNQI-Cu alloy system, $[(\text{DMe})_{1-x}(\text{MeBr})_x\text{-DCNQI}]_2\text{Cu}$ in the small x region. This observation opens a way to study the nature of the reentrant resistivity anomaly at ambient pressure. Detailed studies are now in progress.

References

- 1) A. Aumuller, P. Erk, G. Kerbe, S. Hunig, J. U. von Schutz, and H. -P. Werner, *Angew. Chem., Int. Ed. Engl.*, **25**, 740 (1986).
- 2) A. Kobayashi, R. Kato, H. Kobayashi, T. Mori, and H. Inokuchi, *Solid State Commun.*, **64**, 45 (1987); *Synthetic Metals*, **27**, B275 (1988).
- 3) R. Kato, H. Kobayashi, and A. Kobayashi, *J. Am. Chem. Soc.*, **111**, 5224 (1989).
- 4) T. Mori, K. Imaeda, R. Kato, A. Kobayashi, H. Kobayashi, and H. Inokuchi, *J. Phys. Soc. Jpn.*, **56**, 3429 (1987).
- 5) S. Tomic, D. Jerome, A. Aumuller, P. Erk, S. Hunig, and J. U. von Schutz, *J. Phys. C* **21**, L203 (1988).
- 6) H. Kobayashi, A. Miyamoto, R. Kato, A. Kobayashi, Y. Nishio, K. Kajita, and W. Sasaki, *Solid State Commun.*, to be published.
- 7) The X-ray intensity data were collected on a Rigaku automated diffractometer and corrected for absorption. The final R-values were 0.04-0.08 % (for 13 crystals). A good agreement between the x -value determined by X-ray structure refinement and that by X-ray microanalysis (XMA) has been confirmed in the similar alloy system $\beta\text{-(BEDT-TTF)}_2(\text{I}_3)_{1-x}(\text{AuI}_2)_x$ (A. Kobayashi et al. (to be published)). The details of the structure refinement will be published elsewhere.

(Received July 27, 1989)